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# Copolymerization of Styrene and [Poly(ethylene glycol) methyl ether] Methacrylate using Reverse Atom Transfer Radical Polymerization

Lamya Zahir,<sup>a</sup> Nargish Jahan Ara<sup>b</sup> and Samiul Islam Chowdhury<sup>\*b</sup>

<sup>a</sup>Department of Apparel Engineering, Bangladesh University of Textiles, Tejgaon, Dhaka-1208, Bangladesh <sup>b</sup>Department of Chemistry, Faculty of Science and Engineering, Bangladesh University of Textiles, Tejgaon, Dhaka-1208, Bangladesh

\*Corresponding author E-mail address: sami butex13@yahoo.com (S.I. Chowdhury)

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**Abstract:** Well-defined amphiphilic copolymers of styrene and poly(ethylene glycol) methyl ether methacrylate (PEGMA) were successfully synthesized using reverse atom transfer radical polymerization (RATRP). The initial feed ratio of the macromonomer, PEGMA has a significant effect on the copolymer yield. Copolymer yield decreased with increasing the monomer feed ratio of PEGMA, but number average molecular weight followed the opposite trend. The resultant copolymers were characterized by nuclear magnetic resonance (NMR) spectroscopy to confirm their structures and comonomer compositions. The macromonomer content in the copolymer has a significant effect on the thermal and mechanical properties.

Keywords: Styrene; Copolymers; Macromonomer; RATRP; Mechanical properties

# 1. Introduction

Atom transfer radical polymerization (ATRP) is one of the most promising approaches for controlled radical polymerization.<sup>[1]</sup> ATRP can be used in two different initiating systems: (a) an alkyl halide and transition- metal compound in its lower oxidation state of CuBr complexed by two molecules of ligand for example 4,4'-di(alkyl)-2,2'molecule of bipyridine or one N,N,N',N",N"pentamethyldiethylenetriamine<sup>[2-4]</sup> (b) a conventional radical initiator such as azobis(isobutyronitrile) (AIBN) with the transition-metal compound in its higher oxidation state of CuBr<sub>2</sub> complexed by an appropriate ligand.<sup>[5]</sup> The latter approach has been named reverse atom transfer radical polymerization (RATRP) and was successfully used for copper-based heterogeneous and homogeneous systems in solution and in emulsion as well as for iron complexes.<sup>[5,6]</sup> These systems successfully used to prepare polymers with predetermined molecular weights, low polydispersity (PDI), and precise end functionalities as well as a variety of copolymers with controlled topologies and compositions.<sup>[2,3]</sup>

Styrene is one of the most common monomer among the monomers used in polymerization.<sup>[7]</sup> Polystyrene (PSt) is not only important in academic research but also in industrial research

because it may exhibit excellent properties such as good heat and chemical resistance, high decomposition temperature, high optical transparency and low birefringence.<sup>[8]</sup> The PSt is normally brittle and carried high glass transition temperature ( $T_g$ ).

The properties as well as  $T_g$  value can be controlled using suitable co-monomers composition. The copolymers of styrene (St) can be obtained from different of polymerization system such as coordination,<sup>[9]</sup> anionic<sup>[10]</sup> and controlled radical polymerization (CRP).<sup>[11]</sup> The presence of soft segment for example ethylene,  $\alpha$ olefines and methacrylate derivatives in PSt chain shows excellent properties with controlled  $T_g$  values.<sup>[12-16]</sup> CRP is the most convenient way to copolymerize St with polar monomer such as methacrylate derivatives because of catalytic poisoning or side reaction occurred in coordination and anionic polymerization.

Acrylic-type macromonomers combine the productivity of radical polymerization with high molecular weight polymers with easily tunable degradation time,<sup>[17]</sup> charge density,<sup>[18]</sup> lower critical solution temperature (LCST),<sup>[19]</sup> etc. Poly ethylene glycol (PEG)-based polymers are of interest for a broad range of applications such as biocompatible coatings for magnetic resonance tomography contrast agents,<sup>[20]</sup> drug delivery systems,<sup>[21]</sup> surfactants,<sup>[22]</sup> and for the synthesis of bioconjugates.<sup>[23]</sup> Polymers with short PEG side chains



are applied as dental soft lining materials  $^{\left[24\right]}$  and as polymeric gel electrolytes in lithium batteries.  $^{\left[25\right]}$ 

Copolymerization of styrene and macromonomer is particularly attractive because of well-defined graft properties efficiently impart onto the polymer chain. Furthermore, these macromolecules may exhibit stimuli-responsive properties, which are typically not attainable with linear polymer. For instance, these graft polymers generally display a LCST in pure water or in physiological medium.<sup>[26]</sup> Nonlinear poly(ethylene glycol) methyl ether methacrylate (PEGMA) analogues should appear as ideal structures, which combine both the properties of PEGMA and St in a single macromolecule. Therefore, we are interested to introduce a well-defined PEGMA branch chain into the PSt backbone for tuning the thermal and mechanical properties. In the present paper, we report the copolymerization of St with a macromonomer of PEGMA using RATRP system which is one of the most convenient polymerization method for copolymerization of macromonomers.<sup>[27]</sup>

# 2. Experimental Section

#### 2.1. General considerations

All manipulations were carried out under a nitrogen atmosphere using standard Schlenk-line techniques. All solvents were distilled from sodium/benzophenone or CaH<sub>2</sub>. St (Sigma–Aldrich) was dried over CaH<sub>2</sub> and then freshly distilled under vacuum prior to use. PEGMA ( $M_n$  300 g mol<sup>-1</sup>, Sigma–Aldrich) was activated prior to use by passing through a column with alternating layers of basic aluminum oxide (activated) to remove inhibitor. CuBr<sub>2</sub>, bipyridine, and 2,2'- azobisisobutyronitrile (AIBN) were obtained from Sigma–Aldrich and used as received.

#### 2.2. Analytical procedures

The molecular weight and polydispersity of the polymers were determined by gel-permeation chromatography (GPC) with TOSOH HLC-8320 liquid Chromatograph equipped with an RI detector, using dry THF as the mobile phase. The <sup>1</sup>H and <sup>13</sup>C NMR spectra of the polymers were measured at room temperature on a Bruker 500 MHz spectrometer (Bruker, Germany) operated in pulse Fourier-transform mode. The sample solutions for the <sup>1</sup>H and <sup>13</sup>C NMR measurements were prepared in CDCl<sub>3</sub> with concentrations of up to 5 wt%. The pulse angle was 45°, and approximately 500 scans were accumulated for the  $^{13}\mathrm{C}$  NMR spectra with a pulse-repetition delay of 5.0 s. The peak of residual CHCl<sub>3</sub> in CDCl<sub>3</sub> (<sup>1</sup>H NMR: 7.25 ppm; <sup>13</sup>C NMR: 77.13 ppm) was used as an internal reference. Differential scanning calorimetry (DSC) was performed on an SII EXSTER 600 (Seiko Instruments Inc., Japan) system under a nitrogen atmosphere. Differences in the thermal history of the polymers were eliminated by first heating the specimen to 400 k, cooling it to 200 k, and then recording the second DSC scan (all heating rates: 10 K/min). Thermogravimetric analyses (TGA) were carried out using a TG/DTA 6300 analyzer (Seiko Instruments Inc., Japan) under a flow of nitrogen (constant heating rate: 10 °C/min; 25  $\rightarrow$  450 °C). UV-Vis spectra were recorded on a UV-2600 (Shimadzu Corporation, Japan) system under ambient conditions.

### 2.3. Homo and copolymerization of St and PEGMA

The homopolymerization of St or copolymerization with PEGMA was synthesized as follows: St (5 mmol) [added PEGMA (1 mmol) for copolymerization], were added in 2 mL of toluene into a 10 mL Schlenk equipped with a magnetic bar with five freeze-pump thaw cycles to remove the remaining oxygen. CuBr<sub>2</sub> (11.6 mg, 0.118 mmol) and bpy (36.8 mg, 0.236 mmol) were added quickly to the flask, and one more freeze-pump-thaw cycle was carried out. AIBN (13 mg, 0.059 mmol) dissolved in 1 mL of toluene was then added to the reaction Schlenk. The solution was then degassed with nitrogen, and subsequently placed in a preheated oil bath at 70°C to carry out RATRP polymerization. The polymerization reactions were terminated by adding 50 mL of methanol. The resulting precipitated polymers were collected by filtration, adequately washed with methanol, and dried under reduced pressure.

#### 2.4. Preparation of copolymer film by the solution casting

The copolymers (approximately 300 mg) were dissolved in chloroform (3 mL) at room temperature and filtered through a 0.20  $\mu$ m polytetrafluoroethylene (PTFE) membrane. The filtered solution was placed on a PTFE petri dish and allowed to stand for 3 days at ambient temperature and pressure. After almost all the solvent had evaporated, the film was dried for 3 h under reduced pressure.

## 3. Results and Discussion

#### 3.1. Homopolymerization of St

Homopolymerization of St was conducted by reverse atom transfer radical polymerization (RATRP) using different initiator /catalyst feed ratio at 70 °C in toluene. The results are summarized in Table 1. The highest activity of 95% was achieved in 6h at the ratio of AIBN/CuBr<sub>2</sub>/bpy (1/2/4). This condition was not only provided the highest activity but also gave the polymer with the highest molecular weight with very low PDI ( $\approx$  1.1) than the other initiator-catalyst feed ratio (Table 1, run 3).

#### 3.2. Copolymerization St and PEGMA

Considering the best condition for St Polymerization, copolymerization of St with PEGMA was also carried out using the same initiator/catalyst feed ratio. Such polymer showed branched structure composed of a carbon–carbon backbone of St and multiple oligo(ethylene glycol) side-chains of PEGMA (Scheme 1).

Copolymerization of St/PEGMA showed good conversion at all monomer feed ratios but needed longer polymerization time than the homopolymerization of St. As a result, lower conversion was observed in the copolymerization than the homopolymerization



Scheme 1. Copolymerization of St and PEGMA.



Run	Mono. Ratio	[AIBN]₀/	Time	Conv.	M <sup>b</sup>	PDI	St cont. <sup>c</sup>	T <sub>g</sub> d
	(minol/minol/		(11)	(70)				
1	St (5)	1/0/0	6	97	9118	2.36	100	-
2	St (5)	1/1/2	6	93	7993	2.13	100	-
3	St (5)	1/2/4	6	95	10017	1.11	100	371
4	St/PEGMA (10.0/1)	1/2/4	24	91	20144	1.38	91	297
5	St/PEGMA (7.5/1)	1/2/4	24	88	23169	1.45	87	275
6	St/PEGMA (5.0/1)	1/2/4	24	83	27534	1.59	83	262
7	St/PEGMA (7.5/2)	1/2/4	24	75	29423	1.64	77	255
8	St/PEGMA (7.5/3)	1/2/4	24	63	32735	1.89	70	231

<sup>a</sup>Polymerization condition: Temperature = 70 °C. <sup>b</sup>Determined by size exclusion chromatography (GPC). <sup>c</sup>The content of St in the St/PEGMA copolymer determining by <sup>1</sup>H NMR spectrum. <sup>d</sup>Determined by DSC.



norization of Ct and DECNAA maaromonomor <sup>a</sup>



of St. The co-monomer significantly affects the catalytic activity and the copolymerization yield decreased with increasing the PGMEA ratio from 1 to 3 mmol in monomer feed (Fig. 1). The highest conversion of 91 % was obtained with the lowest co-monomer feed ratio of 10.0/1 [St/PEGMA] (Table 1, Run 4).

The molecular weight of St/PEGMA copolymer was measured by GPC. The increase of PEGMA concentration from 1 to 3 mmol in monomer feed caused the increase in the molecular weights of the produced polymers (Fig. 2) and the PDI was also increased from 1.4 to 1.8 (Table 1, Run 4-8). The highest  $M_n$  of 32735 g/mol was

observed with highest PEGMA ratio with 3 mmol. Unimodal distribution (PDI < 2) indicates that the successful copolymerization.

The incorporation of St in the produced copolymers was investigated by <sup>1</sup>H NMR in CDCl<sub>3</sub>. A typical <sup>1</sup>H NMR spectrum of the PSt and St/PEGMA copolymer (Run 7) is illustrated in Fig. 3. The signals attributed to the aromatic protons of the St Units ( $H^{1-4}$ ) are observed in the range of 6.5–7.1 ppm. The signals assignable to the aliphatic protons of the St and PEGMA units ( $H^{5,6}$  and  $H^7$ ) are observed in the range of 1.2–2.1 ppm. Particularly, the signal of the methylene (-OCH<sub>2</sub>) and methyl (-OCH<sub>3</sub>) proton of the PEGMA unit can



Fig. 3. <sup>1</sup>H NMR spectrum of St (Table 1, Run 3) and <sup>1</sup>H NMR spectrum and <sup>13</sup>C NMR spectra of St/PEGMA copolymer obtained by Run 7 (CDCl<sub>3</sub>, 25 °C, 500 MHz).



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be observed at 3.6-3.75 and 3.4 ppm. The incorporation of St decreased with decreasing St Concentration in the monomer feed ratio (Table 1, Run 4-8). The highest St Content in the St/PEGMA copolymer was achieved to be 91% at the feed ratio of 10:1 (St/PEGMA) (Table 1, Run 4).

A typical <sup>13</sup>C NMR spectrum of the copolymer (Run 7) is shown in Fig. 3. The signals of each chemical shift region were assigned as follows: 127.9 ppm for C<sup>1</sup>, 128.5 and 128.9 ppm for C<sup>2,3</sup>, 143.7 for C<sup>4</sup>, 40.5 and 30.3 ppm for C<sup>5</sup> and C<sup>6</sup> of the St segment, 20.1 ppm for C<sup>7</sup>, 177.7 ppm for C<sup>8</sup>, 71 and 71.8 ppm for C<sup>9,10</sup>, and 59.3 ppm for C<sup>11</sup> of the PEGMA unit. A clear observation of the phenyl-carbons of the St Units also indicates the presence of the St Units in the St/PEGMA copolymer and confirms the random distribution of PEGMA unit in the obtained St/PEGMA copolymer.

#### 3.3. Thermal properties

The thermal properties of the St/PEGMA copolymers were investigated by thermogravimetry-differential thermal analysis (TG-DTA) and differential scanning calorimetry (DSC). The degradation temperature with 5% weight loss of PSt was obtained at 330 °C. The degradation temperature was decreased with increasing the branching density of PEGMA. The degradation temperature (5% weight loss) of St/PEGMA copolymer with 91 mol% St was obtained at 321 °C that was very close to the PSt but the values was decreased to 293 °C for 70 mol% St (Fig. 4). The corresponding DTA curves showed endothermic peak and suggesting the thermal stability of the copolymer.

The  $T_g$  value of the copolymer of St/PEGMA was varied according to the PEGMA content. The PSt showed higher  $T_g$  value as high as 371 K. The  $T_g$  value decreased from 321 to 297 K with only 7 mol % of PEGMA content. According to the PEGMA content, the  $T_g$  value gradually decreased with increasing the PEGMA content and the value observed from 297 to 231 K (Table 2, Run 3-8; Fig. 5).

#### 3.4. Optical property of the St/PEGMA copolymer

PSt is rather brittle because it has poor impact strength due to the stiffness of the polymer backbone and lack of crystallinity.<sup>[28]</sup> Increasing the branching density using PEGMA decreased the brittleness and increased the flexibility in physical appearance of the copolymer film because copolymer with the highest PEGMA content (Table 1, Run 8) only showed gummy product. The light transmittance was measured for the St/PEGMA copolymer thin film

with the thickness about 120  $\mu m.$  The copolymer showed the transmittance around 85% (Table 1, Run 6) in the visible light region (400–800 nm) which is close to the PSt film.  $^{[29]}$ 

# 4. Conclusions

Well-defined amphiphilic St/PEGMA copolymers were obtained by adopting the RATRP method. The highest conversion was observed at the lowest feed ratio of macromonomer PEGMA. On the other hand, the  $M_n$  was increased and PDI was broadening with the highest PEGMA ratio in monomer feed. Thermal and mechanical properties were controlled by the PEGMA content. The copolymer with high content of PEGMA showed lowest  $T_g$  value with high flexibility.

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# **Conflicts of Interest**

The authors declare no conflict of interest

### References

- 1 Matyjaszewski K.; Xia J. Atom Transfer Radical Polymerization. *Chem. Rev.*, 2001, **101**, 2921-2990. [CrossRef]
- 2 Chowdhury S.I.; Hasan T. Synthesis of Well-defined Dihydroxyl End-Functional Polystyrene using Trifunctional Initiator via Atom Transfer Radical Polymerization. *AIJRSTEM*, 2015, 9, 113-118. [CrossRef]
- 3 Chowdhury S.I.; Ali R.; Hasan T. Synthesis of Well-Defined Vinyl End-Functional Polystyrene Using Multifunctional Initiator by Atom Transfer Radical Polymerization. Am. J. Appl. Sci., 2015, 12, 581-587. [CrossRef]
- 4 Chowdhury S.I.; Hasan T. Effect of Spacer Length from Vinyl Group of Vinyl-bromoester Initiator on Atom Transfer Radical Polymerization of Styrene. *AIJRSTEM*, 2015, 9, 212-215. [CrossRef]
- 5 Zhang W.; Yang J.; Li B.; Li X.; Dan Y. Reverse Atom Transfer Radical Polymerization of Styrene in Microemulsions Catalyzed by CuBr<sub>2</sub>/Bpy. *Polym. Sci. Ser. B*, 2013, **55**, 551–555. [CrossRef]
- 6 Fazekas E.; Nichol G.S.; Garden J.A.; Shaver M.P. Iron III Half Salen Catalysts for Atom Transfer Radical and Ring Opening Polymerizations. *ACS Omega*, 2018, **3**, 16945-16953. [CrossRef]
- 7 Chowdhury S.I.; Zahir L.; Hasan T. Kinetic Studies on Bulk Atom Transfer Radical Polymerization of Styrene. *Am. J. Polym. Sci. Technol.*, 2017, **3**, 103-107. [CrossRef]



- 8 Nielsen L.E.; Buchdahl R. Mechanical Properties of Oriented Polystyrene Film. J. Appl. Phys., 1950, **21**, 488-493. [CrossRef]
- 9 Chowdhury S.I.; Tanaka R.; Nakayama Y.; Shiono T. Copolymerization of Norbornene and Styrene with Anilinonaphthoquinone-Ligated Nickel Complexes. *Polymers*, 2019, **11**, 1100. [CrossRef]
- 10 Li Z.; Chen J.; Su L.; Zou B.; Zhan P.; Guan Y.; Zheng A. A Controlled Synthesis Method of Polystyrene-B-Polyisoprene-B-Poly(Methyl Methacrylate) Copolymer via Anionic Polymerization with Trace Amounts of THF Having Potential of a Commercial Scale. *RSC Adv.*, 2017, 7, 9933–9940. [CrossRef]
- 11 Jiang J.; Wang W.-J.; Li B.-G.; Zhu S. Tailoring Uniform Copolymer Composition Distribution via Policy II RAFT Solution Copolymerization of Styrene and Butyl Acrylate. *Macromol. React. Eng.*, 2018, **12**, 1800014. [CrossRef]
- 12 Li S.; Liu D.; Wang Z.; Cui D. Development of Group 3 Catalysts for Alternating Copolymerization of Ethylene and Styrene Derivatives. *ACS Catal.*, 2018, **8**, 6086–6093. [CrossRef]
- 13 Galdi N.; Buonerba A.; Oliva L. Olefin–Styrene Copolymers. *Polymers*, 2016, **8**, 405. [CrossRef]
- 14 Zhang J.; Li J.; Huang L.; Liu Z. Gradient Copolymers of Styrene– Methyl Acrylate and Styrene–Acrylic Acid by Organostibine-Mediated Controlled/Living Radical Polymerization and their Glass Transition Behaviors. *Polym. Chem.*, 2013, **4**, 4639-4647. [CrossRef]
- 15 Shim J.; Bates F.S.; Lodge T.P. Superlattice by Charged Block Copolymer Self-assembly. *Nat. Commun.*, 2019, **10**, 2108. [CrossRef]
- 16 Lutz J.F. Polymerization of Oligo(Ethylene Glycol) (Meth)Acrylates: Toward New Generations of Smart Biocompatible Materials. J. Polym. Sci. Part A: Polym. Chem., 2008, 46, 3459–3470. [CrossRef]
- 17 Yu Y.; Ferrari R.; Lattuada M.; Storti G.; Morbidelli M.; Moscatelli D. PLA-Based Nanoparticles with Tunable Hydrophobicity and Degradation Kinetics. J. Polym. Sci. Part A: Polym. Chem., 2012, 50, 5191–5200. [CrossRef]
- 18 Rooney T.R.; Gumfekar S.P.; Soares J.B.P.; Hutchinson R.A. Cationic Hydrolytically Degradable Flocculants with Enhanced Water Recovery for Oil Sands Tailings Remediation. *Macromol. Mater. Eng.*, 2016, **301**, 1248–1254. [CrossRef]
- 19 Luzon M.; Boyer C.; Peinado C.; Corrales T.; Whittaker M.; Tao L.; Davis T.P. Water-Soluble, Thermoresponsive, Hyperbranched Copolymers Based on PEG-Methacrylates: Synthesis, Characterization, and LCST Behavior. J. Polym. Sci. Part A: Polym. Chem., 2010, 48, 2783-2792. [CrossRef]
- 20 Lutz J.-F.; Stiller S.; Hoth A.; Kaufner L.; Pison U.; Cartier R. One-Pot Synthesis of PEGylated Ultrasmall Iron-Oxide Nanoparticles and their in Vivo Evaluation as Magnetic Resonance Imaging Contrast Agents. *Biomacromolecules*, 2006, 7, 3132-3138. [CrossRef]
- 21 Ulasan M.; Yavuz E.; Bagriacik E.U.; Cengeloglu Y.; Yavuz M.S. Biocompatible Thermoresponsive PEGMA Nanoparticles Crosslinked with Cleavable Disulfide-Based Crosslinker for Dual Drug Release. J. Biomed. Mater. Res. Part A, 2015, **103**, 243-251. [CrossRef]
- 22 Moad G.; Rizzardo E.; Thang S.H. Toward Living Radical Polymerization. *Acc. Chem. Res.*, 2008, **41**, 1133-1142. [CrossRef]
- 23 Börner H.G.; Schlaad H. Bioinspired Functional Block Copolymers. *Soft Matter*, 2007, **3**, 394-409. [CrossRef]
- 24 Wright P.S. Characterization of the Rupture Properties of Denture Soft Lining Materials. J. Dent. Res., 1980, **59**, 614-619. [CrossRef]
- 25 Reiter J.; Michalek J.; Vondrak J.; Chmelikova D.; Pradny M.; Micka Z. Poly(Ethyl Methacrylate) and Poly(2-Ethoxyethyl Methacrylate) based Polymer Gel Electrolytes. J. Power Sources, 2006, 158, 509-517. [CrossRef]
- 26 Lutz J.-F.; Akdemir O.; Hoth A. Point by Point Comparison of Two Thermosensitive Polymers Exhibiting a Similar LCST: Is the Age of Poly(NIPAM) Over? J Am. Chem. Soc., 2006, **128**, 13046–13047. [CrossRef]
- 27 Chowdhury S.I.; Tanaka R.; Nakayama Y.; Shiono T. Synthesis of Norbornene/Divinylbenzene Copolymers Catalyzed by Anilinonaphthoquinone-Ligated Nickel Complexes and their Applications for the Synthesis of Graft Polymers. J. Polym. Sci. Part A: Polym. Chem., 2020, 11. [CrossRef]



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- 28 Li T.; Zhou C.; Jiang M. UV Absorption Spectra of Polystyrene. *Polym. Bull.*, 1991, **25**, 211-216. [CrossRef]
- 29 Worzakowska M. Thermal and Mechanical Properties of Polystyrene Modified with Esters Derivatives of 3-phenylprop-2-en-1-ol. *J. Therm. Anal. Calorim.*, 2015, **121**, 235–243. [CrossRef]